

# Al-Ir-Pd (Aluminum-Iridium-Palladium)

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Recently, [2008Pav] determined the phase relationships in the Al-rich part of this ternary system and presented partial isothermal sections at 1100, 1000, 900, and 790 °C.

## Binary Systems

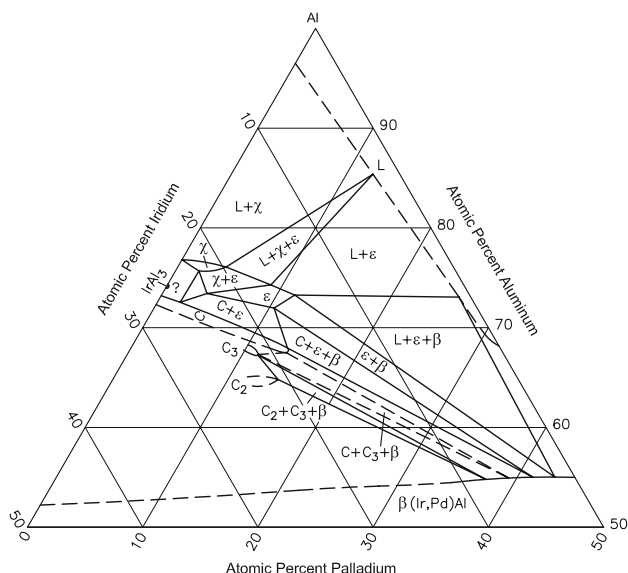
The Al-Ir phase diagram has the following intermediate phases [2008Pav]: Ir<sub>2</sub>Al<sub>9</sub> (Co<sub>2</sub>Al<sub>9</sub>-type monoclinic), Ir<sub>13</sub>Al<sub>45</sub> ( $\phi$ ) (orthorhombic, space group *Pnma*), Ir<sub>9</sub>Al<sub>28</sub> ( $\chi$ ) (hexagonal, space group *P31c*), IrAl<sub>3</sub> (*D0*<sub>18</sub>, Na<sub>3</sub>As-type hexagonal), IrAl<sub>2.7</sub> (cubic, space group *P23*, denoted C), and IrAl (*B2*, CsCl-type cubic, denoted  $\beta$ ). The Al-Pd phase diagram [2001Yur] shows the following intermediate phases: PdAl<sub>4</sub> (hexagonal, space group *P6322*, denoted  $\lambda$ ), PdAl<sub>3</sub> (denoted  $\epsilon_6$ , orthorhombic),  $\epsilon_{28}$  (~PdAl<sub>3</sub>, orthorhombic), Pd<sub>8</sub>Al<sub>21</sub> (Pt<sub>8</sub>Al<sub>21</sub>-type tetragonal), Pd<sub>2</sub>Al<sub>3</sub> (denoted  $\delta$ , *D5*<sub>13</sub>, Ni<sub>2</sub>Al<sub>3</sub>-type hexagonal), PdAl (*B2*, CsCl-type cubic and two low-temperature forms: rhombohedral and *B20*-type cubic), Pd<sub>5</sub>Al<sub>3</sub> (Rh<sub>5</sub>Ge<sub>3</sub>-type orthorhombic), Pd<sub>2</sub>Al (*C23*, Co<sub>2</sub>Si-type orthorhombic), and Pd<sub>5</sub>Al<sub>2</sub> (Pd<sub>5</sub>Ga<sub>2</sub>-type orthorhombic). In the Ir-Pd system [1991Tri], Ir and Pd form a continuous face centered cubic (fcc) solid solution. Below 1480 °C, a miscibility gap occurs in the solid state.

## Ternary Isothermal Sections

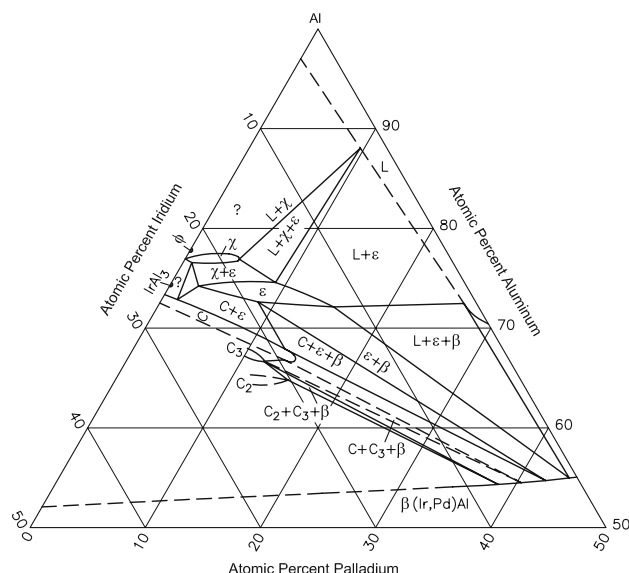
[2008Pav] induction-melted Al-rich compositions under Ar atm and annealed them at 1100-790 °C for

24-2610 h, followed by water quenching. The phase equilibria were studied by x-ray powder diffraction and scanning/transmission electron microscopy. Compositions of co-existing phases were determined by energy dispersive x-ray analysis. Single phase compositions were determined by inductively-coupled plasma optical emission spectroscopy. Differential thermal analysis was carried out at heating/cooling rates of 10-50 °C per min. Partial isothermal sections constructed by [2008Pav] at 1100, 1000, 900, and 790 °C are redrawn in Fig. 1 to 4.

At 1100 °C (Fig. 1), all Al-Pd binary phases except PdAl ( $\beta$ ) are molten. The maximum Al concentration PdAl was ~55 at.%. On the Al-Ir side, Ir<sub>9</sub>Al<sub>28</sub> ( $\chi$ ) dissolves up to 3.5 at.% Pd. IrAl<sub>2.7</sub> (C) dissolves up to 14 at.% Pd. Two ternary phases C<sub>2</sub> and C<sub>3</sub> related to the C phase are present. C<sub>2</sub> (cubic) has a lattice parameter  $a = 1.5482$  nm, which is about twice that of the C phase and has a compositional range around ~Al<sub>65</sub>Pd<sub>13</sub>Ir<sub>22</sub>. The C<sub>2</sub> composition of Al<sub>65</sub>Pd<sub>11.5</sub>Ir<sub>23.5</sub> melts at ~1500 °C. C<sub>3</sub> is hexagonal and has a composition of Al<sub>67</sub>Pd<sub>11.5</sub>Ir<sub>21.5</sub> at 1100 and 1000 °C and lattice parameters of  $a = 1.09135$  nm and  $c = 1.3418$  nm. The full compositional ranges of C<sub>2</sub> and C<sub>3</sub> were not determined. The  $\epsilon$ -related phases were denoted together as  $\epsilon$  by [2008Pav]. This phase occurs in the ternary region between Al<sub>73</sub>Pd<sub>4</sub>Ir<sub>23</sub> and Al<sub>73</sub>Pd<sub>11.5</sub>Ir<sub>15.5</sub>. The binary phase IrAl<sub>3</sub> was not observed in the ternary region. At 1000 °C (Fig. 2), the phase distribution is the same as at 1100 °C. The solubility of Pd in the  $\chi$  phase is 5 at.%. The  $\epsilon$ -range is now wider and extends from

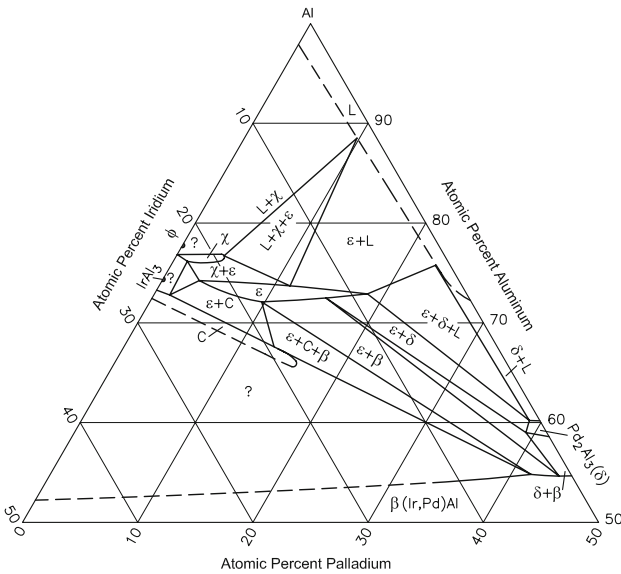


**Fig. 1** Al-Ir-Pd partial isothermal section at 1100 °C for Al-rich alloys [2008Pav]



**Fig. 2** Al-Ir-Pd partial isothermal section at 1000 °C for Al-rich alloys [2008Pav]

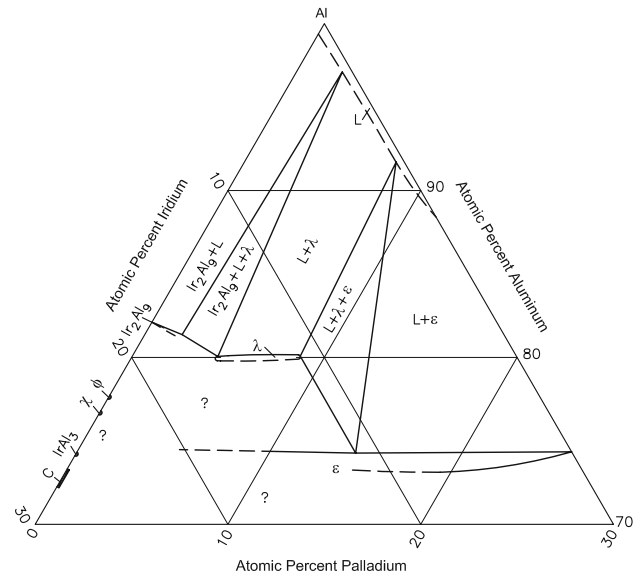
## Section II: Phase Diagram Evaluations



**Fig. 3** Al-Ir-Pd partial isothermal section at 900 °C for Al-rich alloys [2008Pav]

$\text{Al}_{74}\text{Pd}_3\text{Ir}_{23}$  to  $\text{Al}_{73}\text{Pd}_{15.5}\text{Ir}_{11.5}$ . The solubility of Pd in the C phase goes up to 15 at.%. The homogeneity ranges of  $\text{C}_2$  and  $\text{C}_3$  phases extend up to 15.5 and 12 at.% Pd respectively.

At 900 °C (Fig. 3), in addition to the phases at 1100 and 1000 °C,  $\text{Pd}_2\text{Al}_3$  ( $\delta$ ) forms and dissolves up to 2 at.% Ir. The binary phases  $\text{IrAl}_3$  and  $\phi$  were not observed in the ternary region. The solubility of Pd in the  $\chi$  phase decreases to 3 at.%. The  $\varepsilon$ -range extends from  $\text{Al}_{74}\text{Pd}_{3.5}\text{Ir}_{22.5}$  to  $\text{Al}_{73}\text{Pd}_{18.5}\text{Ir}_{8.5}$ . The solubility of Pd in the C phase is at least 13 at.%. At 790 °C (Fig. 4), the  $\varepsilon$  phase extends to the Al-Pd side. The binary phase  $\text{PdAl}_4$  ( $\lambda$ ) appears in the ternary region and has a homogeneity range



**Fig. 4** Al-Ir-Pd partial isothermal section at 790 °C for Al-rich alloys [2008Pav]

from  $\sim\text{Al}_{80}\text{Pd}_{4.5}\text{Ir}_{15.5}$  and  $\sim\text{Al}_{80}\text{Pd}_{8.5}\text{Ir}_{11.5}$ .  $\text{Ir}_2\text{Al}_9$  dissolves about 2 at.% Pd.

### References

- 1991Tri:** S.N. Tripathi, S.R. Bharadwaj, and M.S. Chandrasekharaih, The Ir-Pd (Iridium-Palladium) System, *J. Phase Equilib.*, 1991, **12**(5), p 603-605
- 2001Yur:** M. Yurechko, A. Fattah, T. Velikanova, and B. Grushko, A Contribution to the Al-Pd Phase Diagram, *J. Alloys Compd.*, 2001, **329**, p 173-181
- 2008Pav:** D. Pavlyuchkov, B. Grushko, and T.Ya. Velikanova, An Investigation of the Al-Pd-Ir Phase Diagram between 50 and 100 at.% Al, *J. Alloys Compd.*, 2008, **453**, p 191-196